

and trimethylamine dehydrogenase<sup>19</sup>.

Our results show that structural diversity prevails within the NADP-dependent enzyme family, even when function is closely related. On the other hand, it emphasizes the versatility of the  $\alpha/\beta$ -barrel scaffold, which often appears even in functionally unrelated proteins. Indeed, it is becoming evident that the number of stable folds used to achieve biological diversity is limited.

Preliminary clinical studies strongly support the value of aldose reductase inhibitors in the treatment of diabetic complications<sup>2</sup>. Knowledge of the three-dimensional structure of aldose reductase will enable more specific drugs to be designed so that therapy can eventually be improved. □

Received 7 August; accepted 6 November 1991.

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ACKNOWLEDGEMENTS. We thank A. Van Dorsselaer and M. Jaquinod for unpublished sequence data, B. Rees for his contribution to the phase problem and Y. S. Babu (of Biocryst) for discussions.

## Free $R$ value: a novel statistical quantity for assessing the accuracy of crystal structures

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THE determination of macromolecular structure by crystallography involves fitting atomic models to the observed diffraction data<sup>1</sup>. The traditional measure of the quality of this fit, and presumably the accuracy of the model, is the  $R$  value. Despite stereochemical restraints<sup>2</sup>, it is possible to overfit or 'misfit' the diffraction data: an incorrect model can be refined to fairly good  $R$  values as several recent examples have shown<sup>3</sup>. Here I propose a reliable and unbiased indicator of the accuracy of such models. By analogy with the cross-validation method<sup>4,5</sup> of testing statistical models I define a statistical quantity ( $R_T^{\text{free}}$ ) that measures the agreement between observed and computed structure factor amplitudes for a 'test' set of reflections that is omitted in the modelling and refinement process. As examples show, there is a high correlation between  $R_T^{\text{free}}$  and the accuracy of the atomic model phases. This is useful because experimental phase information is usually inaccurate, incomplete or unavailable. I expect that  $R_T^{\text{free}}$  will provide a measure of the information content of recently proposed models of thermal motion and disorder<sup>6–8</sup>, time-averaging<sup>9</sup> and bulk solvent<sup>10</sup>.

The most common measure for the quality of a crystal structure is the  $R$  value<sup>11</sup>,

$$R = \frac{\sum_{h,k,l} \|F_{\text{obs}}(h, k, l) - |k|F_{\text{calc}}(h, k, l)\|}{\sum_{h,k,l} |F_{\text{obs}}(h, k, l)|} \quad (1)$$

where  $h, k, l$  are the reciprocal lattice points of the crystal,  $|F_{\text{obs}}(h, k, l)|$  and  $|F_{\text{calc}}(h, k, l)|$  are the observed and calculated structure factor amplitudes, respectively.  $R$  is closely related to the crystallographic residual<sup>11</sup>

$$R' = \sum_{h,k,l} (|F_{\text{obs}}(h, k, l)| - |k|F_{\text{calc}}(h, k, l))^2 \quad (2)$$

which is a linear function of the negative logarithm of the likelihood of the atomic model assuming that all observations are independent and normally distributed<sup>12</sup>.  $R$  can be made arbitrarily small by increasing the number of model parameters

and subsequent refinement against  $R'$  (ref. 13), that is the diffraction data can be overfit without improvement or even worsening of the information content of the atomic model.

Crystallographic diffraction data are redundant to some degree, for example, refinement of the penicillopepsin crystal structure from *Penicillium janthinellum*<sup>14,15</sup> at 1.8 Å resolution with 50% of the diffraction data randomly omitted only results in a 0.3 Å root-mean-square (r.m.s.) difference to the atomic structure refined against the full data set (Fig. 1). In analogy to cross-validation<sup>4,5</sup> I thus propose to partition a unique set of the observed reflections into a 'test' set  $T$  and a 'working' set  $A$ , that is,  $T$  and  $A$  are disjoint and their conjunction is the full

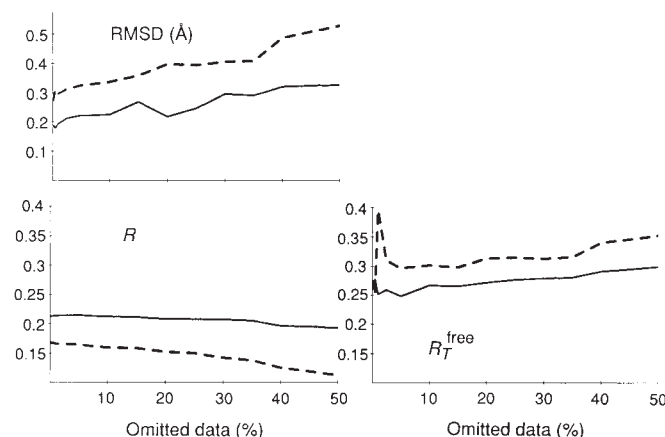


FIG. 1 SA-refinements of penicillopepsin<sup>14,15</sup> at 6–1.8 (solid lines) and 6–2.8 Å (dashed lines) resolution as a function of the percentage of omitted data.  $T$  was obtained by random selection from a unique set of all observed reflections.  $R$  is computed for the  $A$  set of reflections. The r.m.s. differences (RMSD) are computed between the structures refined against  $A$  and a unique set of all observed reflections. The RMSD is unequal to zero for 0% of the data omitted; this reflects the r.m.s. difference between two independently refined structures<sup>25</sup>. The penicillopepsin crystal structure<sup>14,15</sup> without water molecules and unit occupancy values was used as the starting point. Each refinement consisted of a slow-cooling protocol<sup>26</sup> using the program X-PLOR<sup>16,27</sup> starting at 1,000 K, overall B-factor refinement, and restrained individual B-factor refinement with the target values for the temperature factor deviations<sup>2</sup> of 1.5, 2, 2, 2.5 for bonded backbone, angle-related backbone, bonded sidechain, and angle-related sidechain atoms, respectively.

set of observed reflections. I refer to

$$R_T^{\text{free}} = \frac{\sum_{(h,k,l) \in T} \|F_{\text{obs}}(h, k, l) - k|F_{\text{calc}}(h, k, l)\|}{\sum_{(h,k,l) \in T} |F_{\text{obs}}(h, k, l)|} \quad (3)$$

as the free  $R$  value computed for the  $T$  set of reflections.  $T$  is omitted in the modelling process, for example in the case of crystallographic refinement<sup>2</sup> the residual to be minimized is given by

$$R'_A = \sum_{(h,k,l) \in A} (|F_{\text{obs}}(h, k, l) - k|F_{\text{calc}}(h, k, l)|)^2. \quad (4)$$

One would expect that  $R_T^{\text{free}}$  is less prone to overfitting than  $R$ . This concept can be applied to other statistical quantities, such as the standard linear correlation coefficient<sup>11</sup>. It can even be applied to crystal structures which have already been refined with all diffraction data included: refinement by simulated annealing (SA)<sup>16</sup> with  $T$  omitted will remove some of the memory towards  $T$ .

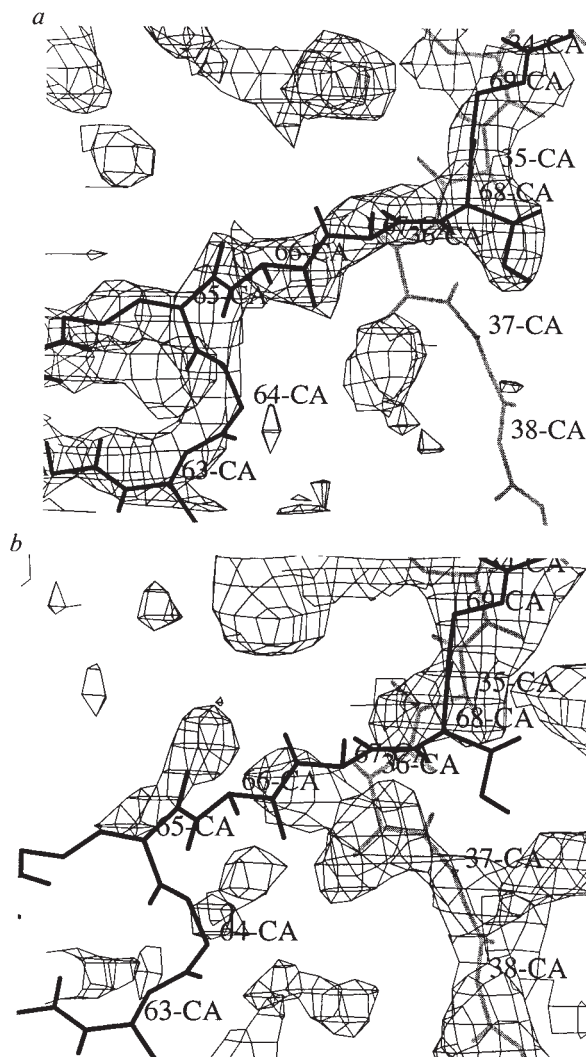
$R_T^{\text{free}}$  reflects the information content of the atomic model. Suppose both the atomic model and diffraction data are perfect, resulting in  $R=0$ . Refinement against  $A$  as opposed to all data will not change the atomic model and thus  $R_T^{\text{free}}=0$ . Suppose the data contain small errors and an atomic model is overfit to a very low  $R$  value by introducing a large number of free parameters. As the noise is independent among different reflections, overfitting against  $A$  will not bias  $R_T^{\text{free}}$ . A similar argument applies to the case of partially incomplete or incorrect atomic

models where the agreement with the diffraction data is improved by fitting noise.

The enhanced sensitivity of  $R_T^{\text{free}}$  with respect to model errors is illustrated in Fig. 2 which compares a portion of the correct<sup>17</sup> and incorrect<sup>18</sup> crystal structures of the plant ribulose-1,5-biphosphate carboxylase oxygenase (RuBisCO). Although the  $R$  difference between the correct and incorrect model is only 4% for comparable geometry, the  $R_T^{\text{free}}$  difference is 13%, suggesting that the incorrect model had been overfit. This is corroborated by the electron density maps in Fig. 2 which show a poorer agreement for the incorrect model. An alternative way to detect the errors in the RuBisCO crystal structure is provided by computing 'omit maps' with simulated annealing (A. Hodel, D. Eisenberg, S.-H. Kim and A.T.B., manuscript in preparation) which essentially is the real-space analogue to  $R_T^{\text{free}}$ .

Both  $R_T^{\text{free}}$  and the r.m.s. difference between the model refined against the complete data set and against  $A$  increase more or less monotonically as a function of the percentage of omitted data (Fig. 1). This is to be expected for terms that truly monitor the validity of a model.  $R$  decreases, which is paradoxical and misleading behaviour for an indicator of the models accuracy. As a compromise between avoiding fluctuations of  $R_T^{\text{free}}$  and maintaining small r.m.s. differences between refined models, I suggest  $T$  is obtained from a random selection of 10% of the observed reflections. The definition of  $R_T^{\text{free}}$  implies  $R_T^{\text{free}} > R$ ; the difference between  $R_T^{\text{free}}$  and  $R$  is uniformly distributed as a function of resolution (not shown).

FIG. 2 The region around residue 66 of the small subunit of RuBisCO. The correct<sup>17</sup> structure is shown in black, whereas the incorrect<sup>18</sup> structure, which involved the nearly backwards tracing of the polypeptide chain of the small subunit, is shown in grey. Superimposed are  $\sigma_A$ -weighted<sup>28</sup>  $2F_o - F_c$  electron density maps with phases computed from the correct model (a,  $R=0.16$ ,  $R_T^{\text{free}}=0.34$ ) and incorrect model (b,  $R=0.2$ ,  $R_T^{\text{free}}=0.47$ ) shown at 2.5 Å resolution for a contour level of  $1\sigma$ . The maps are ordinary omit maps (for a review of omit map techniques, see A. Hodel, D. Eisenberg, S.-H. Kim and A.T.B., manuscript in preparation), that is residues 36–47 and all residues within 5 Å of this loop were removed in the phase calculation.  $T$  was obtained by a 10% random selection from the observed reflections. SA-refinements and restrained B-factor refinements were done at 2.5 Å resolution using  $A$ . The r.m.s. deviations of bond lengths and bond angles from ideal were 0.02 Å and 4°, respectively, for the correct structure whereas they were 0.03 Å and 5° for the incorrect structure.



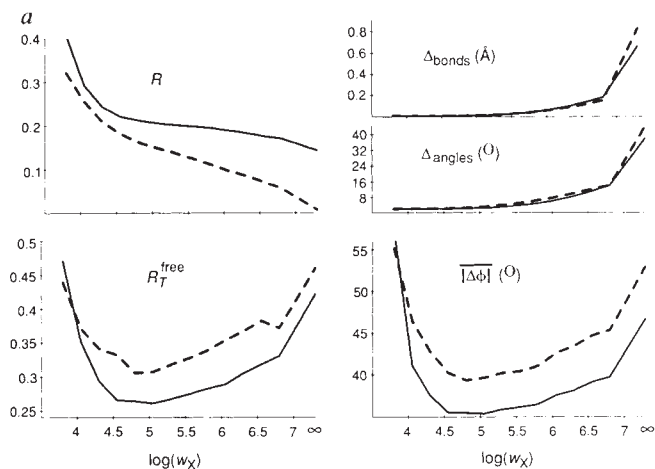
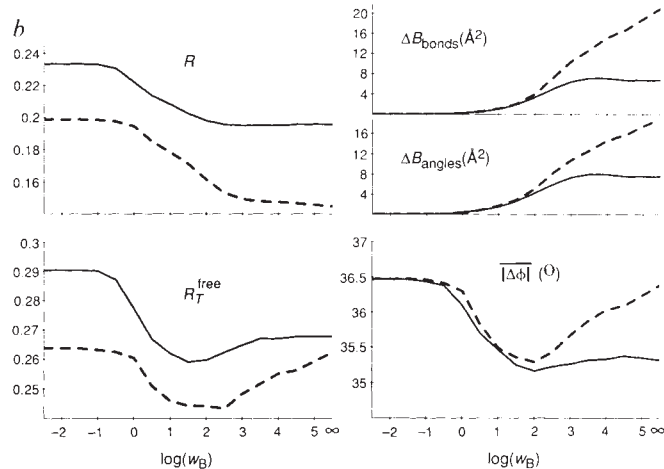


FIG. 3 *a*, SA-refinements of penicillopepsin<sup>14,15</sup> at 6–1.8 (solid lines) and 6–2.8 Å (dashed lines) resolution as a function of  $w_x$  (equation (5)) with  $R'$  replaced by  $R'_A$  (equation (4)).  $T$  was obtained by a 10% random selection.  $R$  was computed for  $A$ .  $\Delta_{\text{bonds}}$  and  $\Delta_{\text{angles}}$  are the r.m.s. deviations of bond lengths and bond angles from their ideal values.  $|\Delta\Phi|$  is the figure-of-merit weighted mean phase difference between model phases and the most probable MIR phases at 6–2.8 Å resolution. Details of the penicillopepsin model and refinement procedure are the same as in Fig. 1. The standard



linear correlation coefficient between the  $R_T^{\text{free}}$  and  $|\Delta\Phi|$  graphs is 0.98 for both resolution ranges. *b*, Restrained B-factor refinements of penicillopepsin as a function of  $w_B$  (equation (6)) with  $R'$  replaced by  $R'_A$  (equation (4)).  $\Delta B_{\text{bonds}}$  and  $\Delta B_{\text{angles}}$  are the r.m.s. deviations between B-factors of atoms sharing a covalent bond or bond angle, respectively.  $w_B = \infty$  represents the completely unrestrained case whereas  $w_B \rightarrow -\infty$  represents refinement of a single overall B-factor.

Diffraction data and prior knowledge are often combined as is the case in restrained least-squares refinement of atomic positions<sup>2</sup> that can be viewed as minimization of the atomic coordinates against a cost function<sup>19,20</sup>

$$C = w_x R' + E_{\text{chemical}} \quad (5)$$

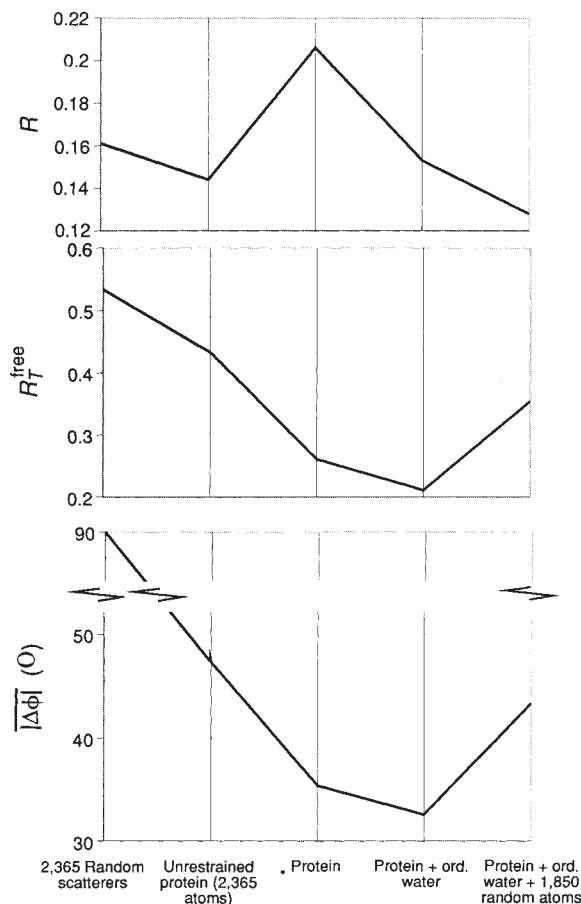
where  $w_x$  is a weight and  $E_{\text{chemical}}$  is a geometric<sup>19</sup> or empirical<sup>21</sup> energy function which has been made unitless by multiplication of a conversion factor. If  $w_x$  chosen is too small, too much emphasis is put on the geometry as provided in  $E_{\text{chemical}}$ , which results in an inaccurate  $R$  value. If  $w_x$  is chosen too large, the structure will be overfit to a very good  $R$  value, but the geometry of the structure becomes severely distorted. The optimal choice of  $w_x$  cannot be obtained by linear hypothesis tests<sup>13</sup> because of the presence of nonlinear restraints, such as repulsive contact functions<sup>2</sup>.  $R_T^{\text{free}}$  is not subject to such limitations.

A series of positional refinements of the penicillopepsin structure<sup>14,15</sup> produced a minimum for  $R_T^{\text{free}}$  at  $\log(w_x) = 5$ , independent of the resolution range used (Fig. 3*a*). At this minimal value the r.m.s. deviation of bond lengths and bond angles from ideality are 0.013 Å and 2.5°, respectively. As an independent determination of the optimal  $w_x$  I used the multiple isomorphous replacement (MIR) phases at 6–2.8 Å resolution<sup>14,15</sup>; these phases were of exceptional quality with a figure of merit of 0.9. Experimental phase information is normally less accurate, incomplete or missing.  $R_T^{\text{free}}$  is highly correlated with the mean difference between the model and the MIR phases ( $|\Delta\Phi|$ ) (Fig. 3*a*).  $R_T^{\text{free}}$  thus yields the optimal choice for  $w_x$  without reference

to experimental phase information or to expected deviations of the geometry from ideality. The resulting relatively tight geometry is a consequence of the diffraction data, not of the geometric or empirical energy function.

Restrained temperature factor refinement<sup>22</sup> poses a similar

FIG. 4 '2,365 Random scatterers' consists of 2,365 oxygen atoms with a reduced van der Waals' radius of 1.57 Å randomly placed in the asymmetric unit of the crystal. 'Unrestrained protein', consists of the same scatterers placed near the non-hydrogen positions of the protein portion of the penicillopepsin structure refined with chemical restraints (equation (5)). 'Protein + ord. water' includes an additional 314 ordered water molecules. 'Protein + ord. water + 1,850 random atoms' includes an additional 1,850 oxygen atoms randomly placed in the bulk solvent region. The definition of  $T$  and  $|\Delta\Phi|$  is identical to Fig. 3. Each refinement consisted of a two iterations of SA-refinement and restrained B-factor refinement as detailed in Fig. 1. The standard linear correlation coefficient between the  $R_T^{\text{free}}$  and  $|\Delta\Phi|$  graphs is 0.98.



problem as positional refinement. It consists of minimization of a cost function

$$C = w_B R' + \sum_{(i,j) \text{ bonds}} \frac{(B_i - B_j)^2}{\sigma_{\text{bonds}}^2} + \sum_{(i,j,k) \text{ angles}} \frac{(B_i - B_k)^2}{\sigma_{\text{angles}}^2} \quad (6)$$

where  $B_i$  is the temperature factor of atom  $i$  and the summations are carried out over all covalent bonds and bond angles<sup>22</sup>.  $R_T^{\text{free}}$  is highly correlated with  $|\Delta\Phi|$  and thus determines the optimal choice of  $w_B$  (Fig. 3b).

The information content of a random distribution of scatterers is obviously minimal, although it can be refined to a very low  $R$  value (Fig. 4);  $R_T^{\text{free}}$  stays at 54% which is close to the random limit of 59% for an acentric space group<sup>11</sup>. Unrestrained refinement with a model consisting of the same scattering starting at the positions of the non-hydrogen protein atoms yields  $R_T^{\text{free}} = 43\%$  (Fig. 4). Thus,  $R_T^{\text{free}}$  can distinguish between a distribution of scatterers that is close to the crystal structure and a random distribution, both of which can be refined to a very low  $R$ . Inclusion of chemical restraints increases  $R$  somewhat while greatly decreasing both  $R_T^{\text{free}}$  and  $|\Delta\Phi|$ , thus improving the information content of the model (Fig. 4). Inclusion of ordered water molecules lowers  $R$ ,  $R_T^{\text{free}}$  and  $|\Delta\Phi|$  (Fig. 4). Refinement of randomly placed scatterers in the bulk solvent region of the crystal lowers  $R$  while increasing both  $R_T^{\text{free}}$  and  $|\Delta\Phi|$ , thus decreasing the information content.

$R_T^{\text{free}}$  represents a reliable and unbiased parameter by which to evaluate the information content of a model produced by X-ray crystallography. It is not restricted to high-resolution diffraction data: tests carried out both at 6–2.8 Å and at 6–1.8 Å resolution produce large correlations between  $R_T^{\text{free}}$  and  $|\Delta\Phi|$  (Fig. 3). The observation that  $R_T^{\text{free}}$  can distinguish between a random distribution of scatterers and distribution close to the protein suggests applications to *ab initio* phasing. The increase of  $R_T^{\text{free}}$  on modelling the bulk solvent region of the penicillopepsin structure with stationary atoms confirms the disordered character of bulk solvent. A similar approach might be useful for the three-dimensional structure determined by solution NMR<sup>23,24,29</sup> if sufficient redundancy in the data and accuracy of the NOE (nuclear Overhauser effect) intensities can be achieved. □

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**Acknowledgements** are brief; grant and contribution numbers are not allowed.

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Received 14 August; accepted 30 October 1991.

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ACKNOWLEDGEMENTS. I thank M. N. G. James and A. R. Sielecki for providing the diffraction data and coordinates of the penicillopepsin structure, D. Eisenberg for providing the data and coordinates of RuBisCo and the Pittsburgh Supercomputer Center for support.

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